

ESTUDO DA ZAC

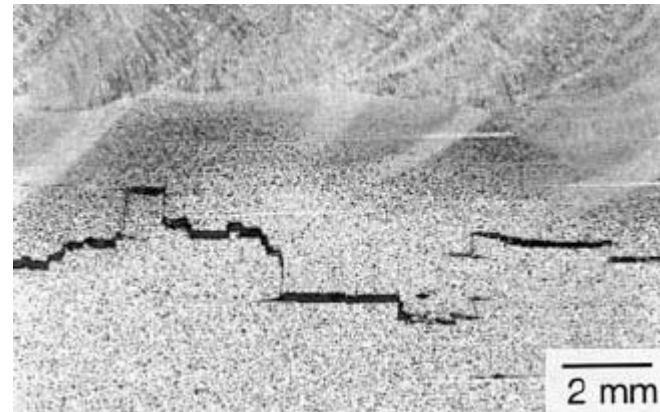
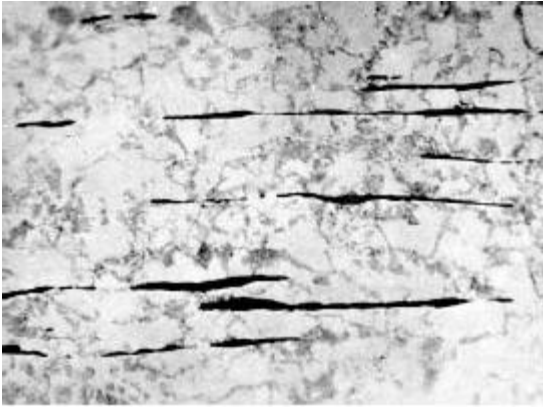
TRINCAS DE REAQUECIMENTO

DECOESÃO LAMELAR

- **S. Kou** / Cap. 17 Transformation-Hardening Materials: Carbon and Alloy Steels
- **Bailey** / Cap. 4 e 6 Lamellar Tearing e Reheat Cracking
- **Mestrados** de Yasunobu Aihara e Alaor R. Amaral / Posmec
- **Artigos** diversos (Hornbogen e Kreye, Tenckhoff etc)

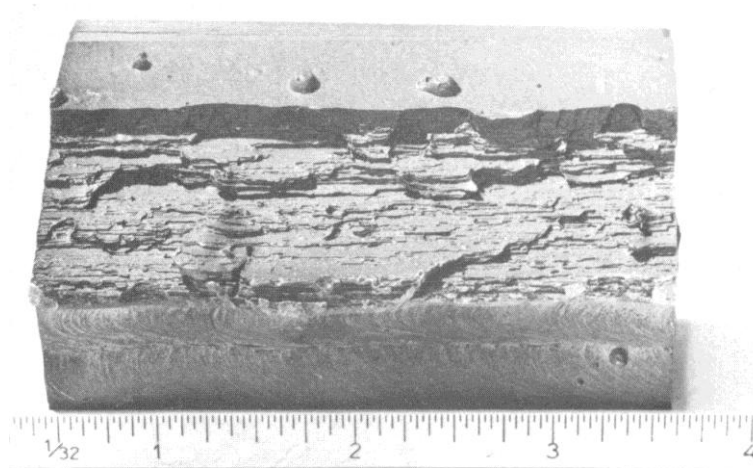
TABLE 17.1 Typical Welding Problems and Practical Solution in Carbon and Alloy Steels, and Their Locations in the Text

Typical Problems	Alloy Types	Solutions	Locations
Porosity	Carbon and low-alloy steels	Add deoxidizers (Al, Ti, Mn) in filler metal	3.2 3.3
Hydrogen cracking	Steels with high carbon equivalent	Use low-hydrogen or austenitic stainless steel electrodes Preheat and postheat	3.2 17.4
Lamellar tearing	Carbon and low-alloy steels	Use joint designs that minimize transverse restraint Butter with a softer layer	17.6
Reheat cracking	Corrosion and heat-resisting steels	Use low heat input ^a to avoid grain growth Minimize restraint and stress concentrations Heat rapidly through critical temperature range, if possible	17.5
Solidification cracking	Carbon and low-alloy steels	Keep proper Mn/S ratio	11.4
Low HAZ toughness due to grain growth	Carbon and low-alloy steels	Use carbide and nitride formers to suppress grain growth Use low heat input ^a	17.2 17.3
Low fusion-zone toughness due to coarse columnar grains	Carbon and low-alloy steels	Grain refining Use multipass welding to refine grains	7.6 17.2

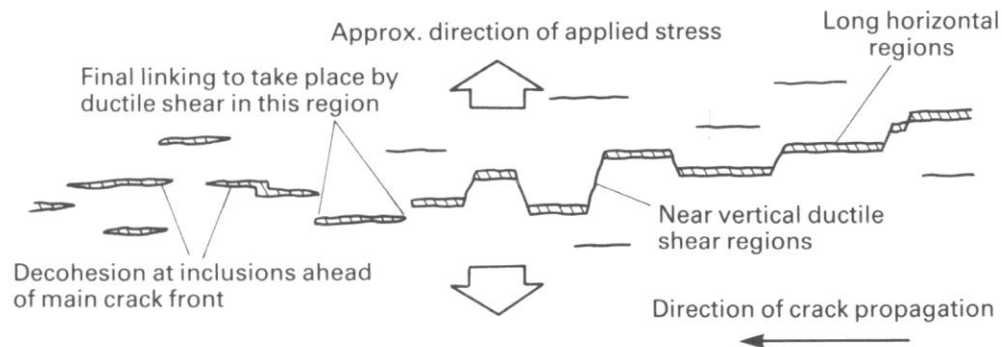


Microstructure of a lamellar-tearing susceptible steel.

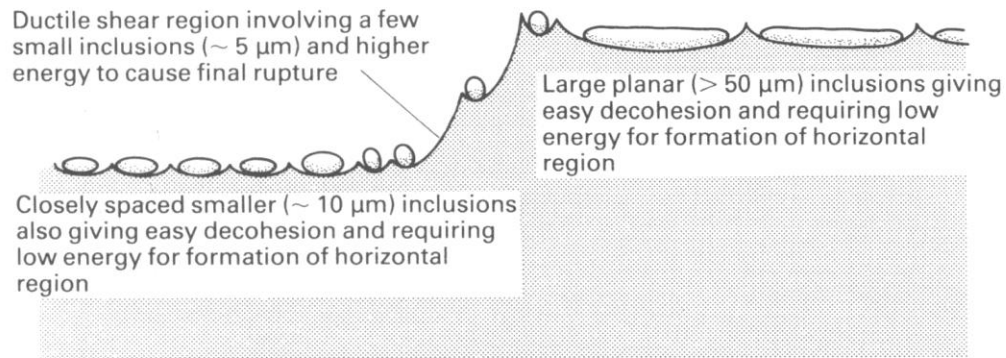
Lamellar tearing near a C-Mn steel weld.



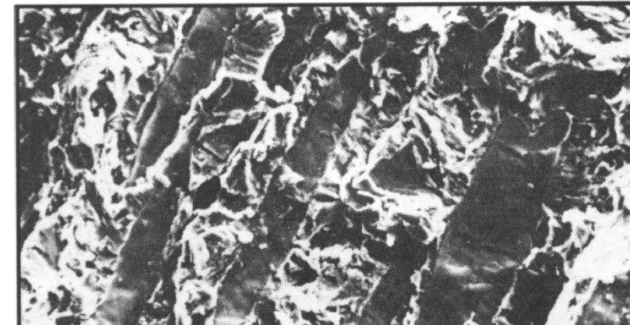
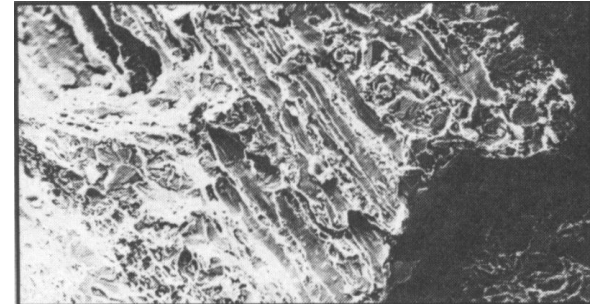
Decoção Lamelar

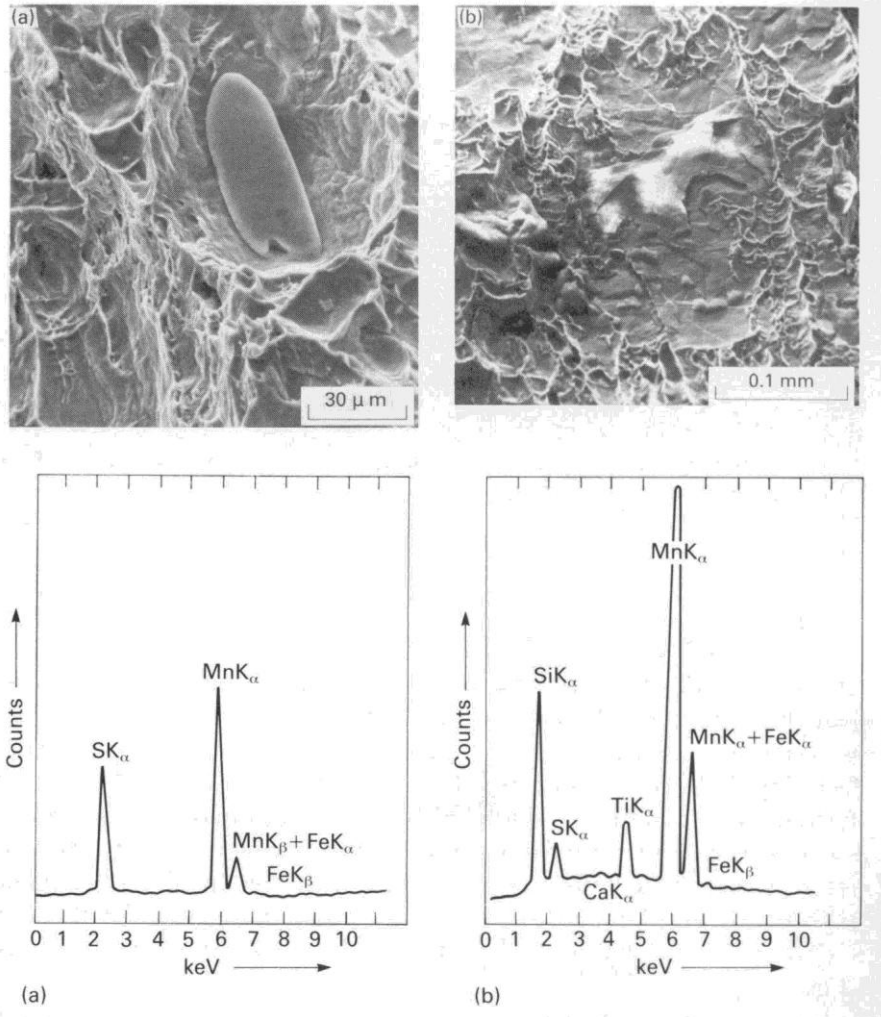


Schematic diagram of a lamellar tear illustrating principal features



Detail of a portion of a lamellar tear





$S > 0.01\%$ provável

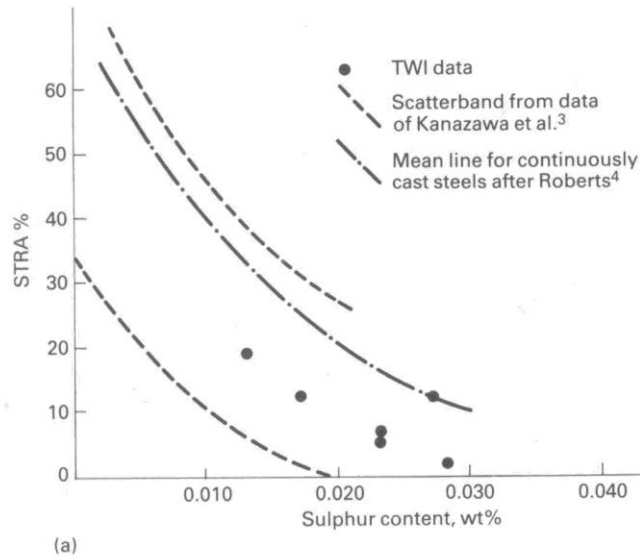
$S < 0,01\%$...baixo risco

$S \sim 0,003\%$...sem risco

Inclusões típicas associadas com decoção lamelar e espectro de EDS

a) Tipo I MnS

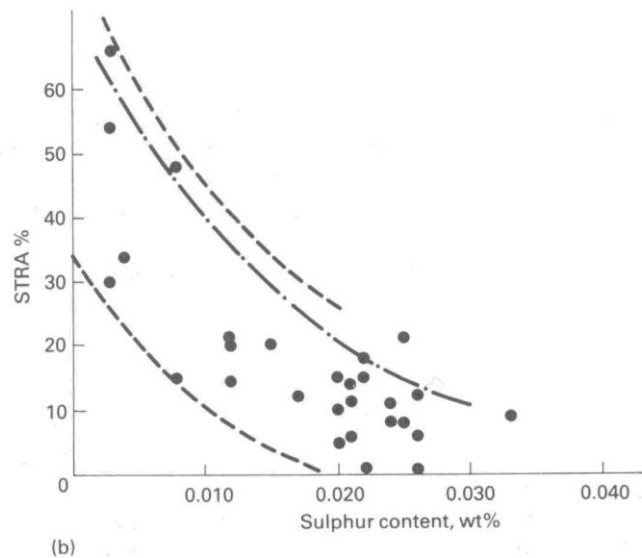
b) silicato complexo

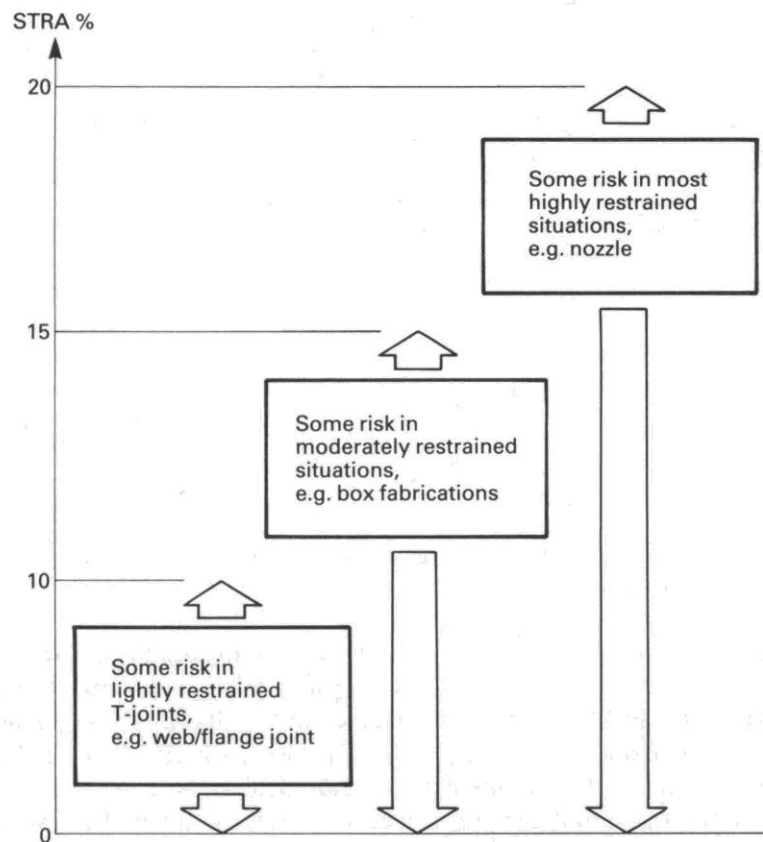


Influência do teor de S no aço base
sobre a estricção em z

a) Chapa com espessura < 12,5 mm

b) Espessura 12,5 a 50 mm

























































































Risco de decoção lamelar em função da estricção em z

DECOESÃO LAMELAR RECOMENDAÇÕES PARA SELEÇÃO DA CLASSE DE AÇO (SEL096)			
RDL	Grau de Qualidade	Estrição em Z	
		Média	Mínimo
até 10	-	-	-
11 a 20	1	15	10
21 a 30	2	25	15
>30	3	35	25

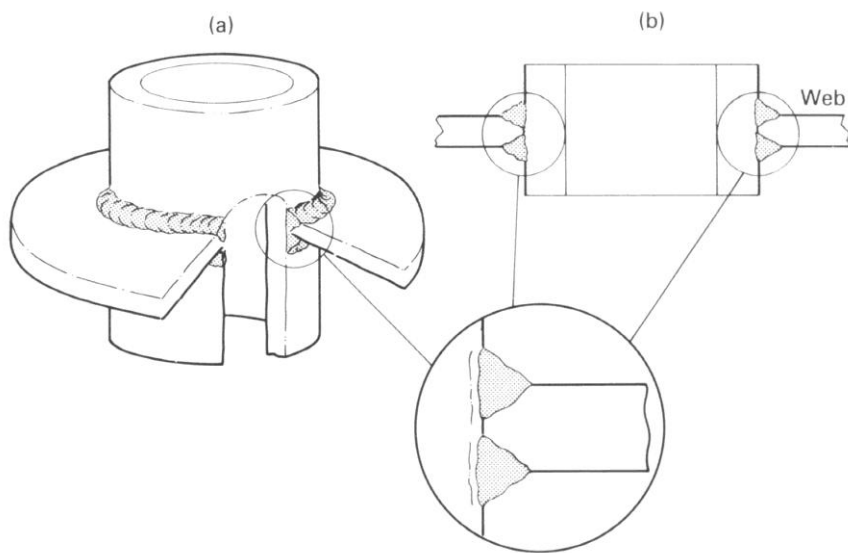
RISCO DE DECOESÃO LAMELAR

$$RDL = INF (A) + INF (B) + + INF (E)$$

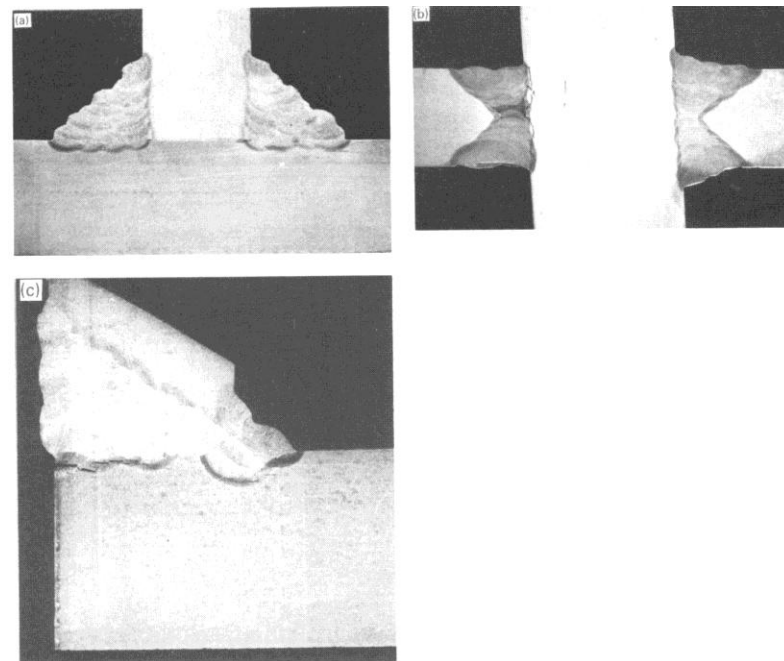
Fator de influência (INF)			
A	Espessura efetivo do cordão. a_d (1)	$a_d \leq 10 \text{ mm}$	3
		$10 < a_d \leq 20 \text{ mm}$	5 (2)
B	Configuração da junta.	$20 < a_d \leq 30 \text{ mm}$	9 (3)
		$30 < a_d \leq 40 \text{ mm}$	12
		$40 < a_d \leq 50 \text{ mm}$	15
C	Espessura da chapa solicitada na direção da espessura. - s -		-25
			
			
			
			
			
			
D	Condições de restrição.		-10
			
			
			
			
			
			
E	Préaquecimento.		-5
			
			
			
			
			
			
E	Préaquecimento.		0
			
			
			
			
			
			
E	Préaquecimento.		3
			
			
			
			
			
			
E	Préaquecimento.		5
			
			
			
			
			
			
E	Préaquecimento.		8
			
			
			
			
			
			
E	Préaquecimento.		2
			4
			6 (3)
			8
			10
			12
			
E	Préaquecimento.		0
			
			
			
			
			
			
E	Préaquecimento.		3
			
			
			
			
			
			
E	Préaquecimento.		5
			
			
			
			
			
			
E	Préaquecimento.		8
			
			
			
			
			
			

Decoção Lamelar

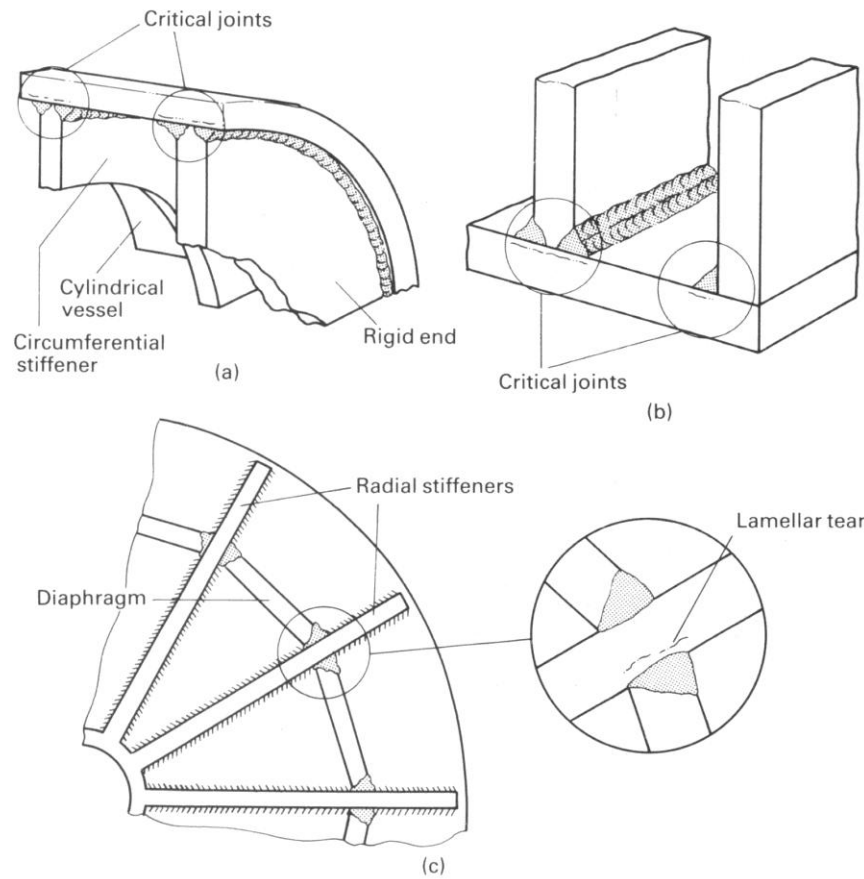
Decoção Lamelar



Juntas soldadas
susceptíveis

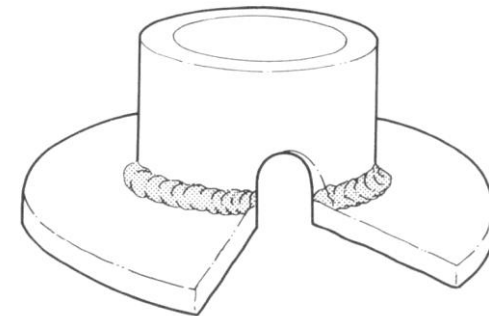
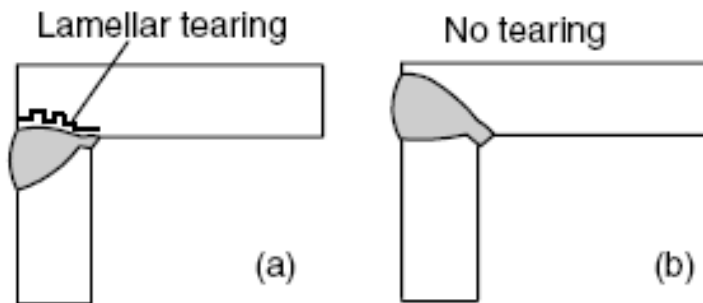


Decoção Lamelar

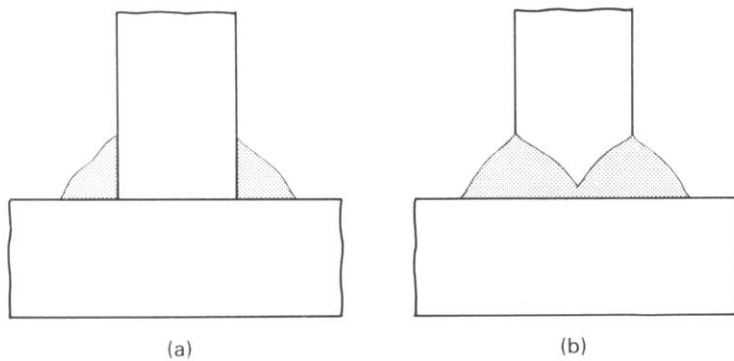


Juntas soldadas
susceptíveis

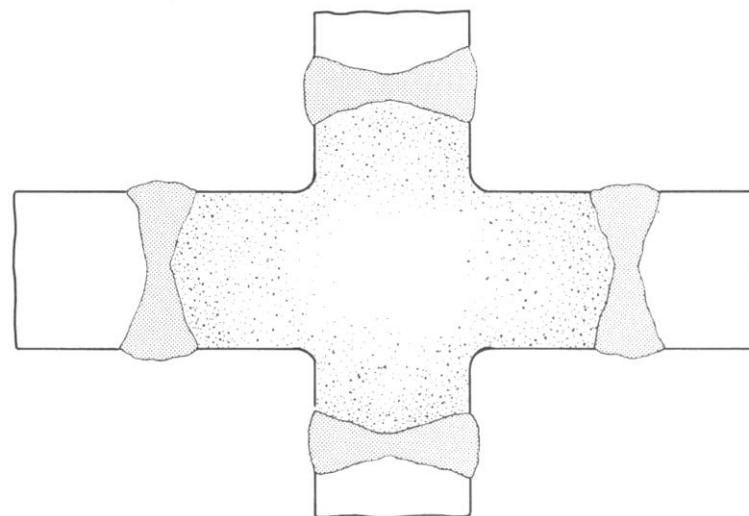
Soluções de Projeto



Lamellar tearing of a corner joint:
(a) *improper design*; (b) *improved design*.

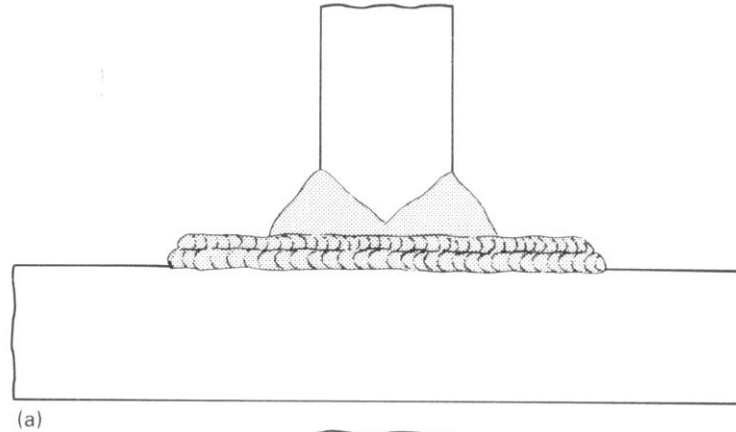


Juntas em T filete (a) menos
susceptíveis que de tôpo (b)

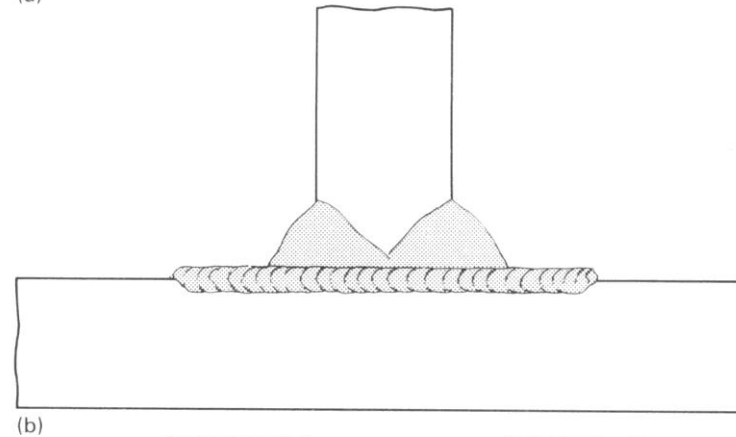


Risco reduzido por:

a) Amanteigamento

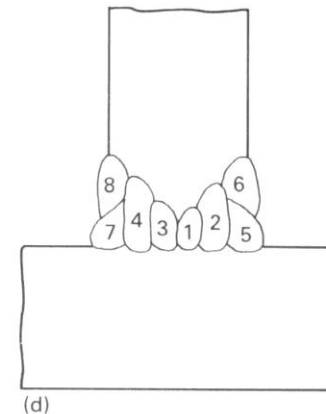
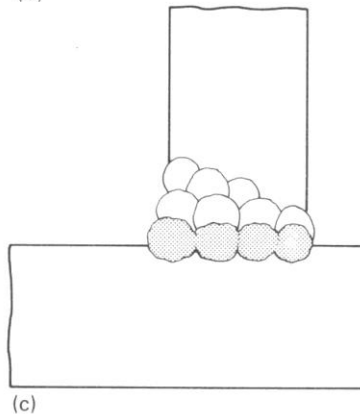


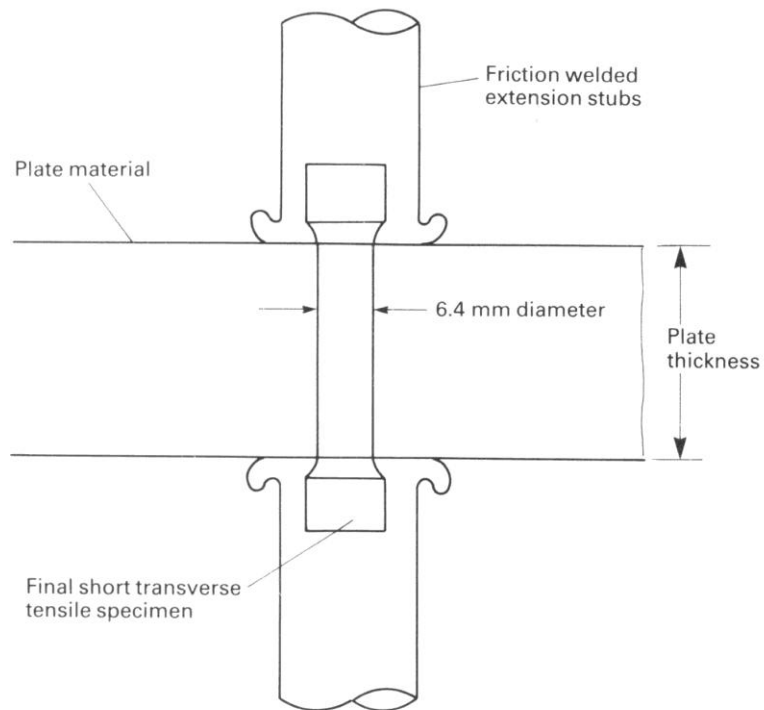
b) Remoção de material susceptível e amanteigando



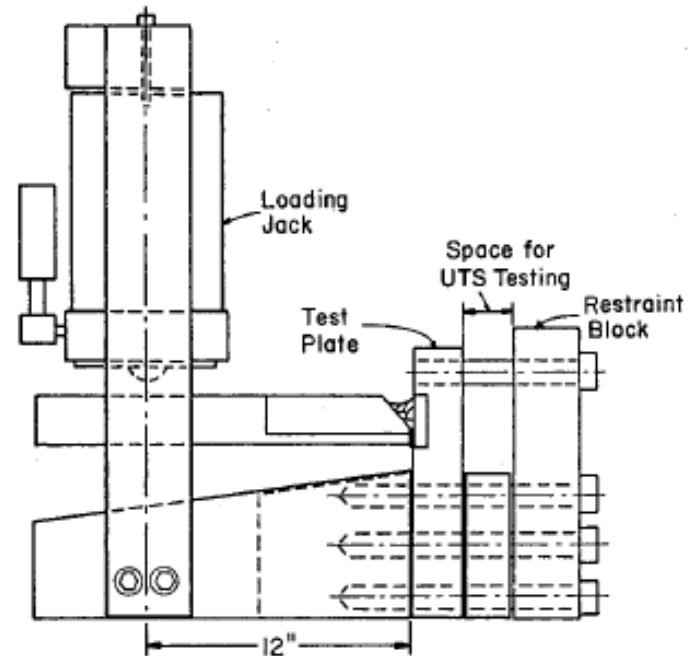
c) Amanteigamento *in situ*

d) *Balanceando sequência de soldagem quando decoção provém de raiz de solda original do tipo (c).*

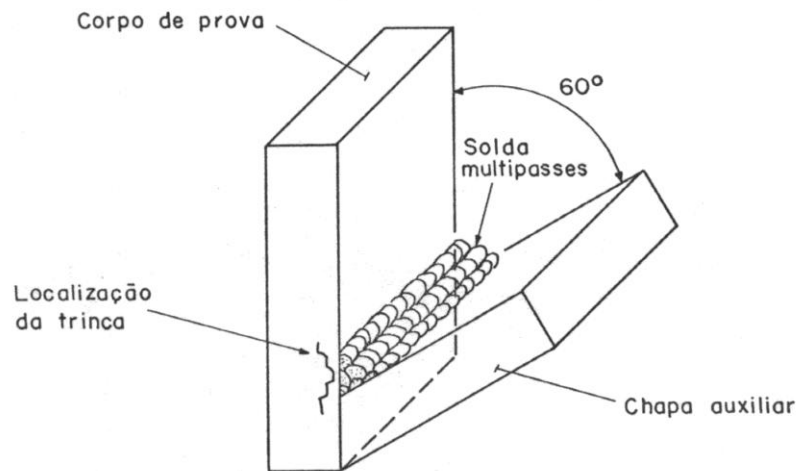




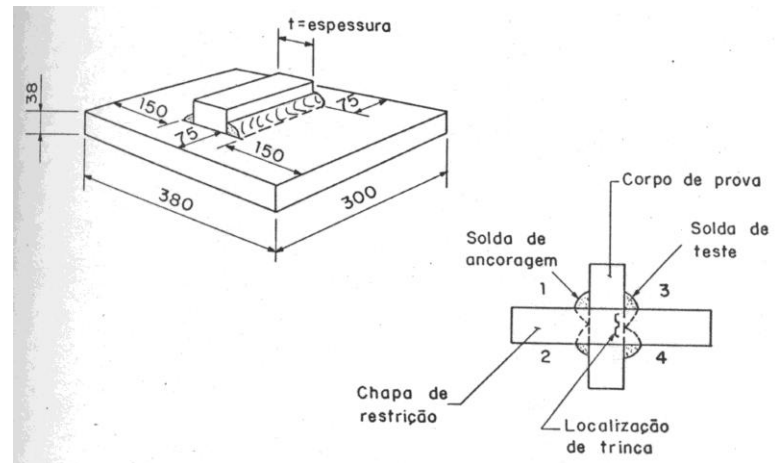
Ensaio Indireto
Tração p/ estrição-z



Ensaio Direto Quantitativo
The Lehigh cantilever lamellar tearing test.

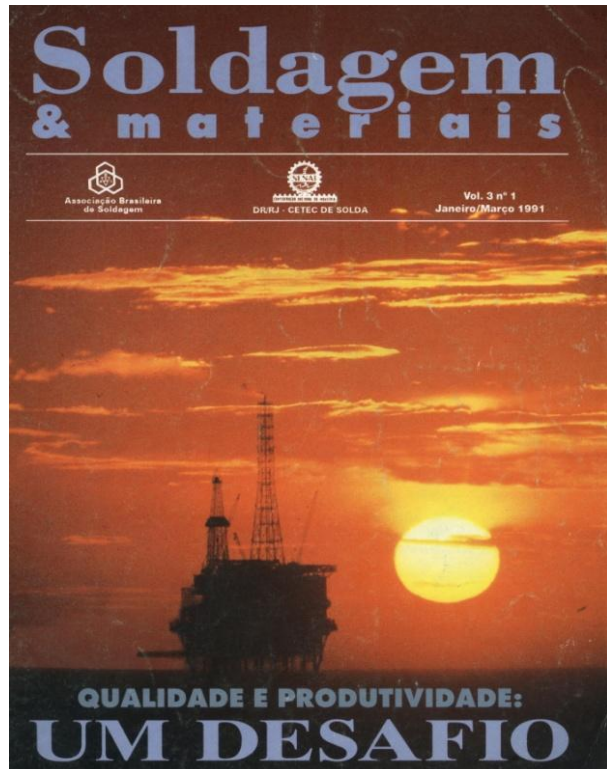


Ensaio de Cranfield



Ensaio da "Janela"

Ensaios Diretos Qualitativos



SUPLEMENTO DE PESQUISA

AVALIAÇÃO QUANTITATIVA DA DECOÇÃO LAMELAR E CRITÉRIO DE AMOSTRAGEM

Y. AIHARA
A. J. A. BUSCHINELLI

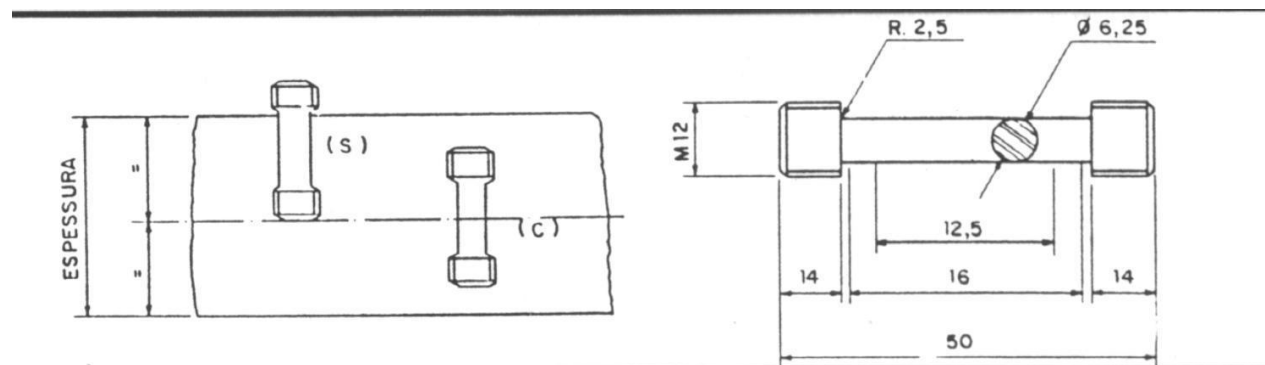
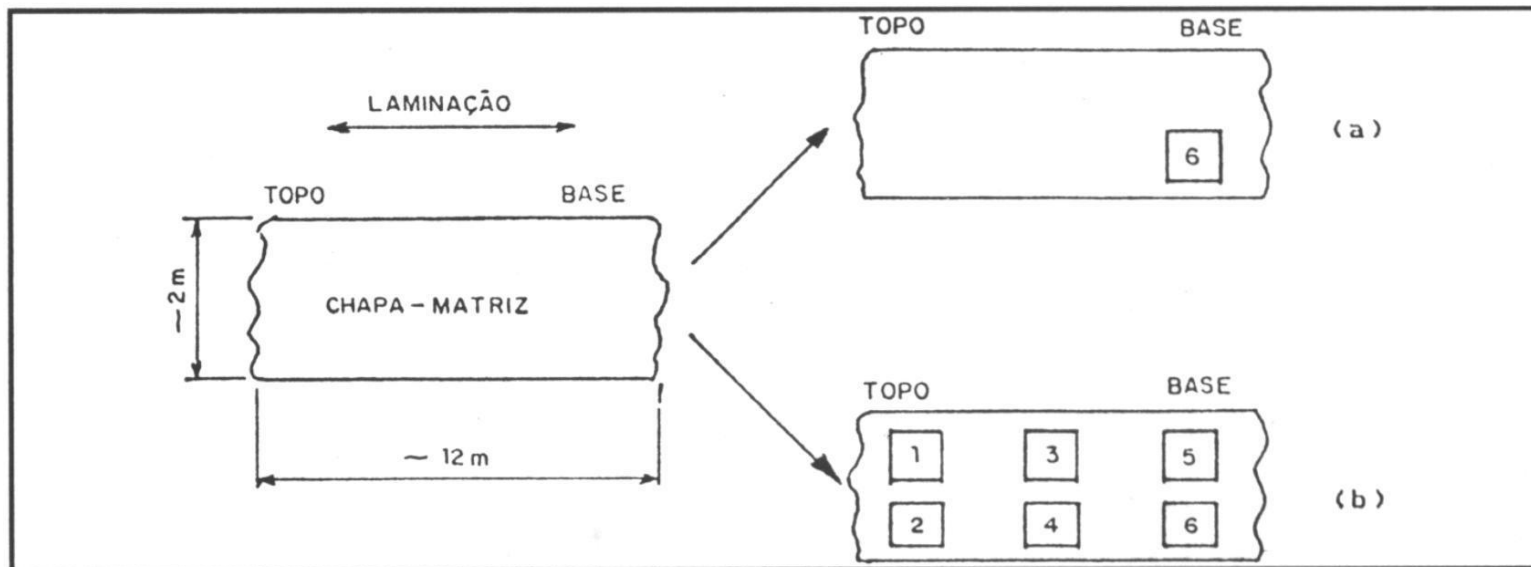
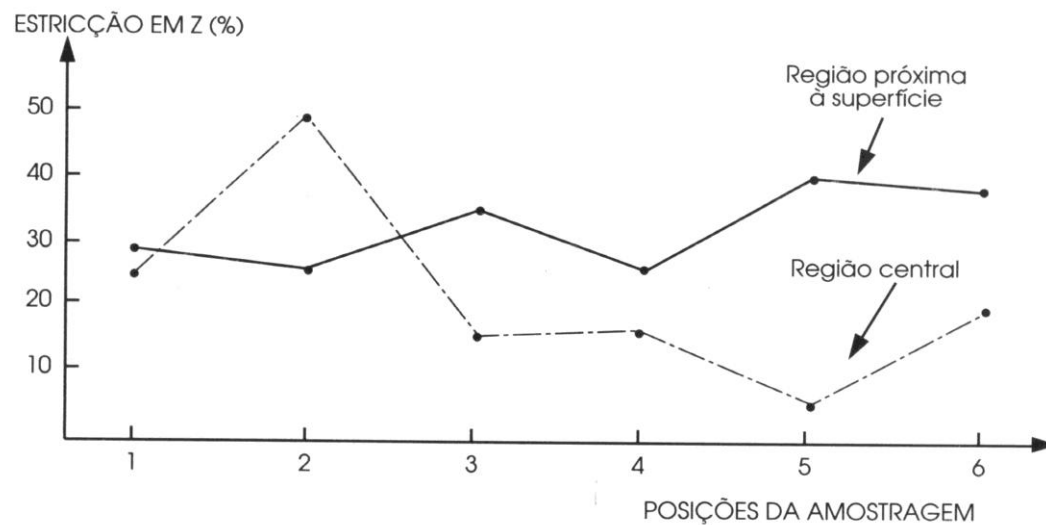
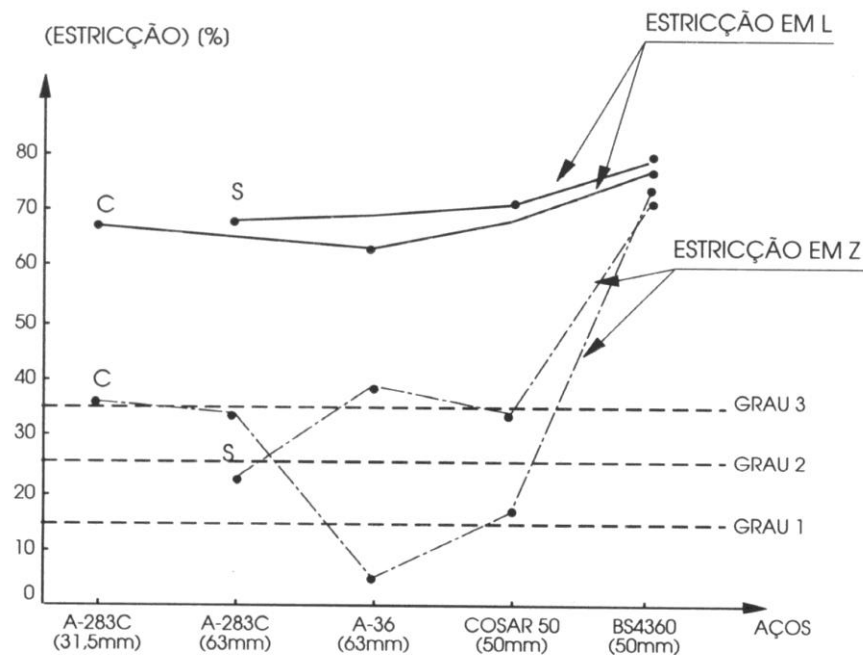


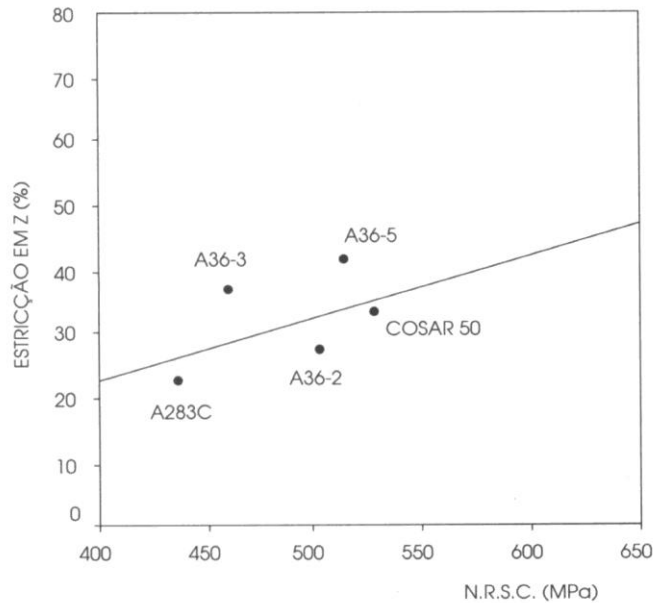
TABELA 1 - Características dos Aços Conforme Fabricante

material	A-283C	A-283C	A-36	COSAR50	BS4360
esp. (mm)	31.5	63	63	50	50
L.E. (MPa)	258	—	328	375	405
LR (MPa)	414	—	498	530	516
A (Lo-50mm)	29	—	32	32	30
laminação	conv.	conv.	con.	control.	conv.
t. térmico	—	—	—	—	normal.
deso (AL-SI)	s.acalm.	s.acalm.	acalmado	acalmado	acalmado

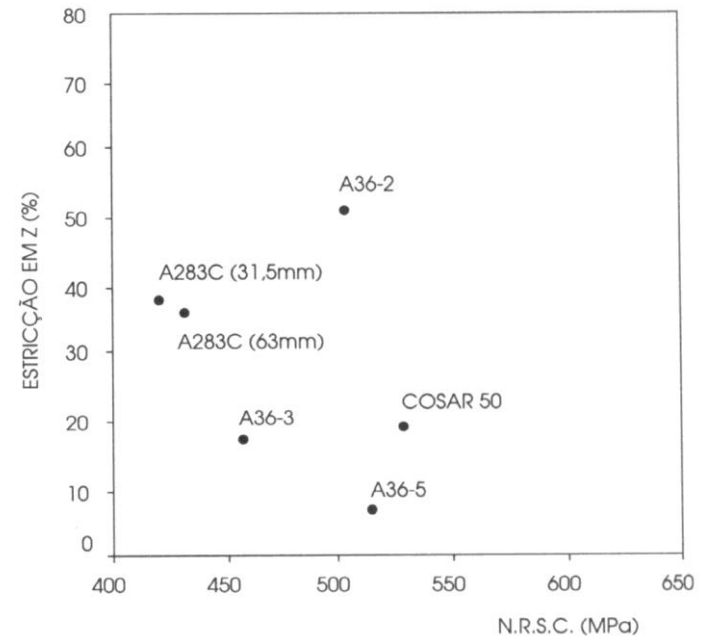
Decoção Lamelar



Correlação resultados da **estricção em z** e
o **NRSC do ensaio de Lehigh**



a) região próxima à superfície



b) Região central da chapa

CONCLUSÕES

1. Excessiva variação da estricção-z ao longo da espessura exige amostragem seletiva → o método do IIW para espessuras maiores que 25mm pode não ser representativo.
2. Existe boa correlação entre o NRSC do ensaio de Lehigh e a estricção-z medida seletivamente junto à superfície da chapa.
3. O local de amostragem na chapa matriz é relevante → para aço acalmado A-36 foram medidas variações da ordem de 900% na estricção-z da região central da espessura!

Histórico

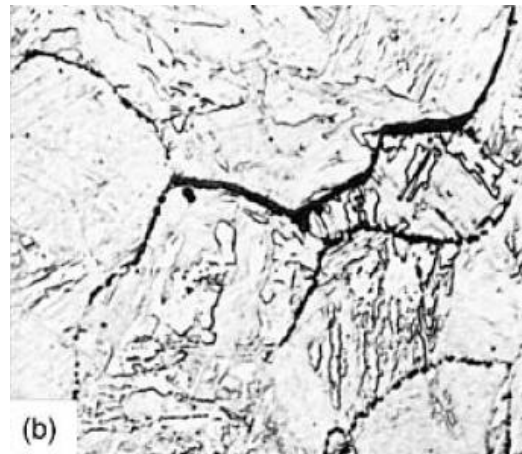
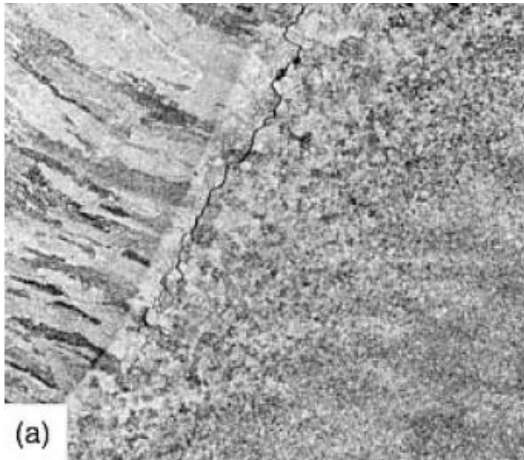
Anos 60 → trincas em tubos de aço AISI 347 em termoeletricas no Canadá

70/80 → trincas sob revestimento inox em vasos de pressão em reatores nucleares PWR

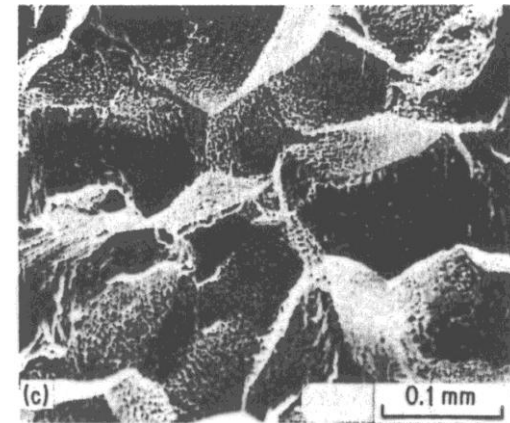
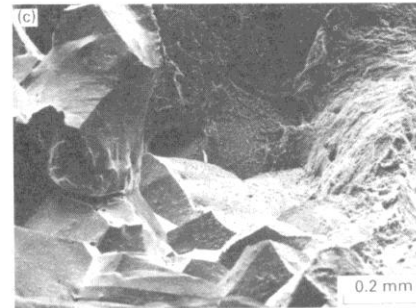
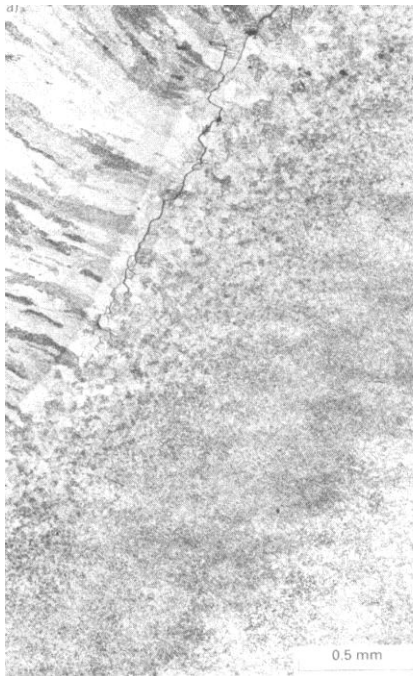
90 → trincas em aços Cr-Mo ($<3\%Cr$)

Características

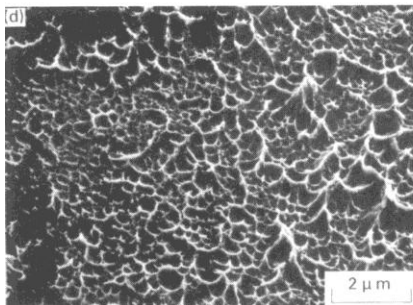
Microtrincas intergranulares na ZAC-GG → risco de fragilização !!!



Trincas de reaquecimento em aço CrMoV: (a) *macroestrutura* (35x); (b) *microestrutura* (1000x).

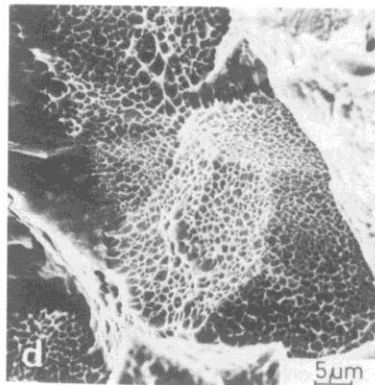
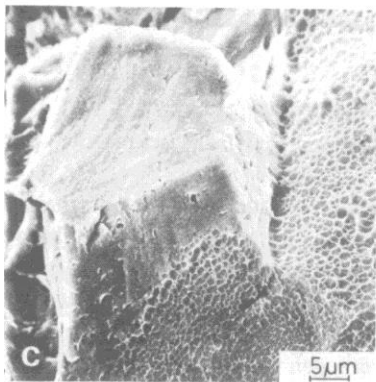
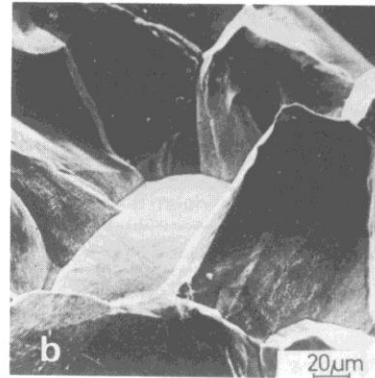
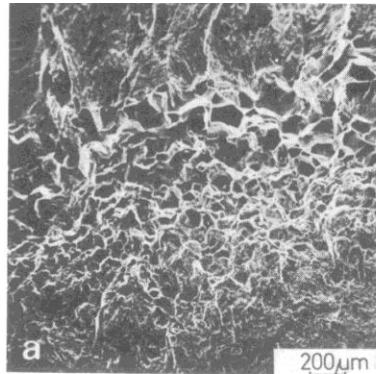


Microdimples indicando **fratura pseudo-intercristalina** *Hornbogen & Kreye*



Aspecto típico das trincas:

- a) macroestrutura mostrando trinca intergranular na ZAC-GG;
- b) MO junto a ponta da trinca revelando cavidades em contornos de grãos;
- c) MEV mostrando fratura intergranular;
- d) microcavidades nas faces dos grãos.



Trinca de alívio de tensões, aspecto da fratura (sob revestimento em aço 22NiMoCr37); a) visão macro; b) ZAC-GG: contornos de grão lisos da austenita original; c) região de transição para ZAC-N; d) grãos menos grosseiros: contornos com cavidades.

Tenckhoff 1979

MECANISMOS DE FRAGILIZAÇÃO

TRINCAS DE REAQUECIMENTOT ~ 550 a 680 oC

- 1) RE-PRECIPITAÇÃO COERENTE NO INTERIOR DOS GRÃOS AUSTENÍTICOS
- 2) FORMAÇÃO DE ZONA LIVRE DE PRECIPITADO ZLP JUNTO AOS CONTORNOS DE GRÃO DA γ ORIGINAL

FRAGILIDADE DE REVENIDOT ~ 500 oC

SEGREGAÇÃO DE IMPUREZAS NOS CONTORNOS DE GRÃO

$$\text{MCF} = \text{Si} + 2 \text{ Cu} + 2 \text{ P} + 10 \text{ As} + 15 \text{ Sn} + 20 \text{ Sb}$$

Grupo VI a \rightarrow S, Se e Te

Grupo V a \rightarrow N, P, As, Sb e Sn

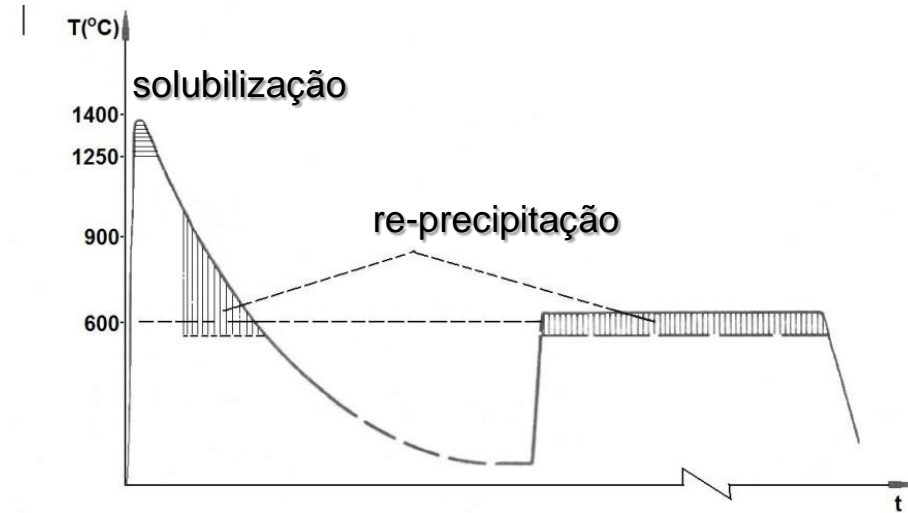
OCORRÊNCIA EXIGE CONJUNÇÃO DE FATORES:

- Reaquecimento na faixa de $T \sim 550$ a 650 oC
- Tensões trativas elevadas \sim LE
- ZAC grosseira
- Microestrutura susceptível \rightarrow martensita ou bainita
- Composição susceptível \rightarrow Mo, V, Nb ...

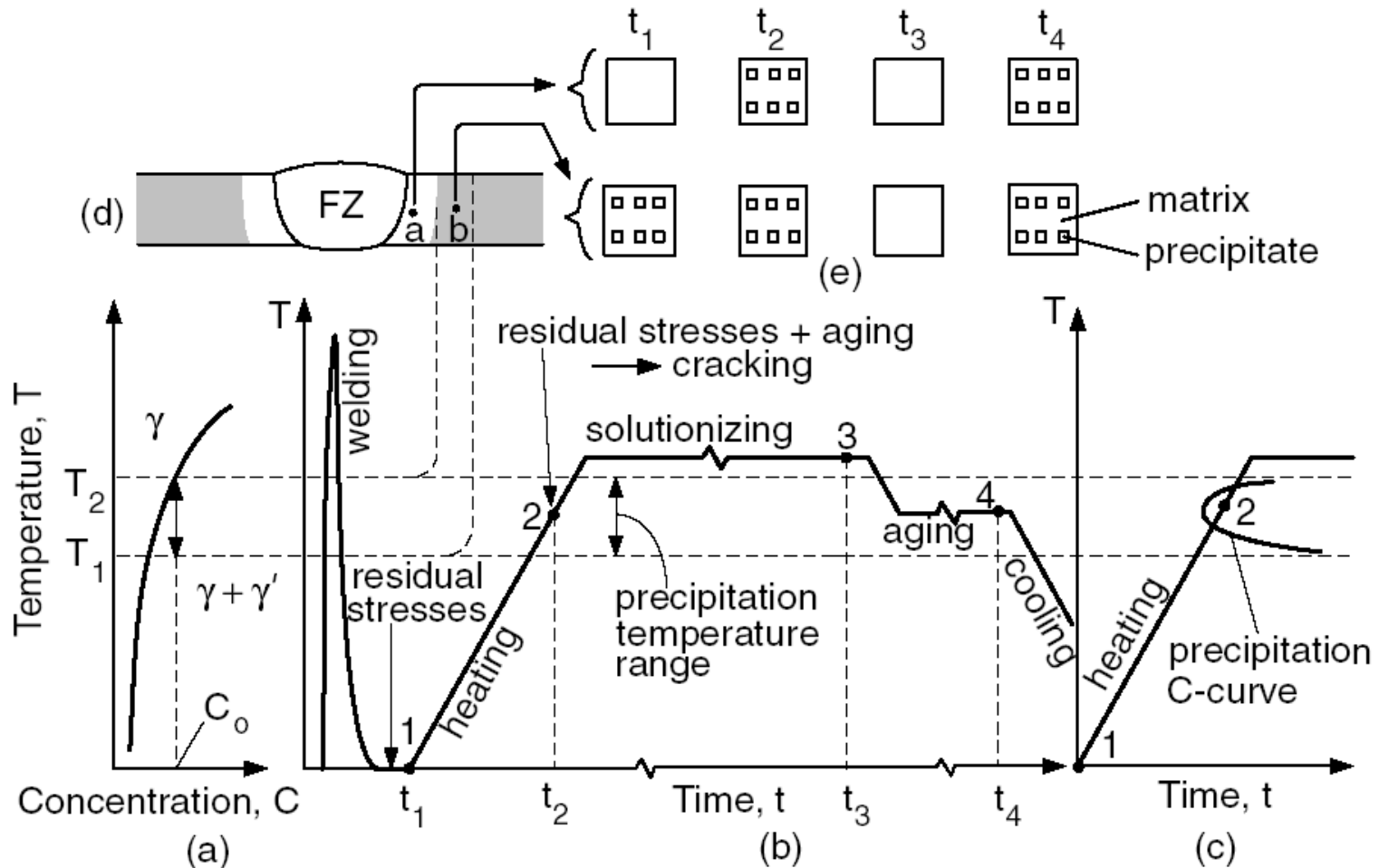
Exemplos de aços sujeitos a trincamento:

0.5Cr–0.5Mo–0.25V
0.5Cr–1Mo–1V
2.25Cr–1Mo

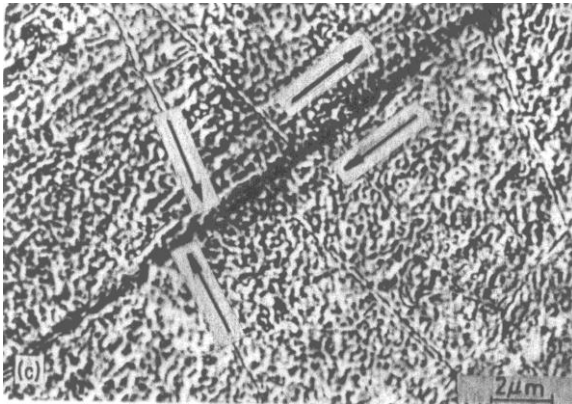
22NiCrV37
AISI 347



Solubilização e precipitação de carbo-nitretos durante a soldagem e TTAT de aços BLAR microligados.



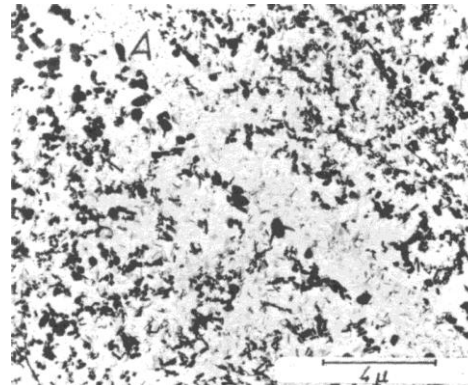
Postweld heat treatment cracking: (a) phase diagram; (b) thermal cycles during welding and heat treating; (c) precipitation C curve; (d) weld cross-section; (e) changes in microstructure.



Deslizamento relativo de grãos austeníticos junto a ponta da trinca.

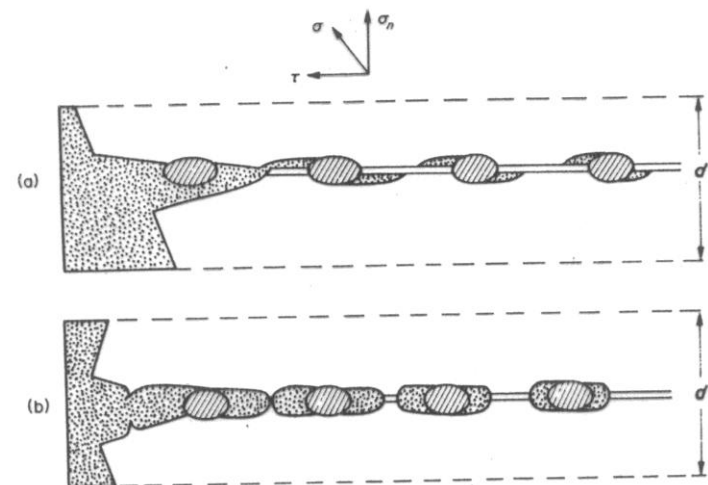


TEM de ZLP em Al3%Cu,
300 oC / 16 min *Hornbogen & Kreye*



ZLP de carbonitreto em
aço StE 36 (1300oC/6' /
H2O + 600oC/1h / H2O)

Brenner & Kreye 1977

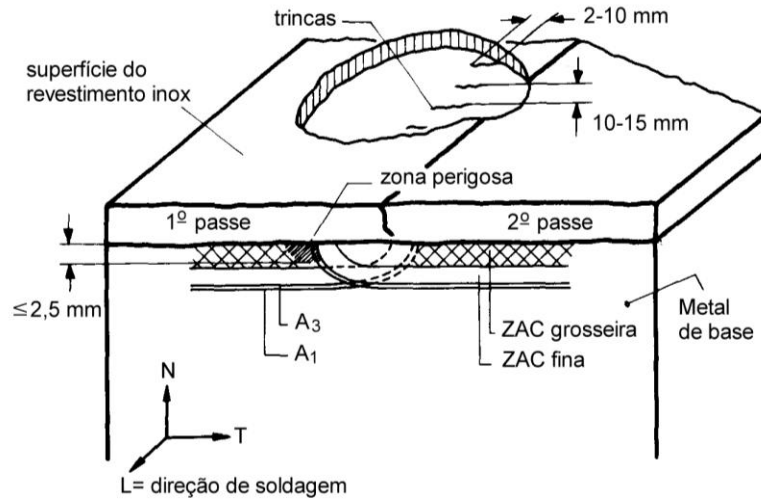


Mecanismo de separação na ZLP por formação de poros junto a precipitados incoerentes causada por (a) tensão cizalhante e (b) tração.

Hornbogen & Kreye

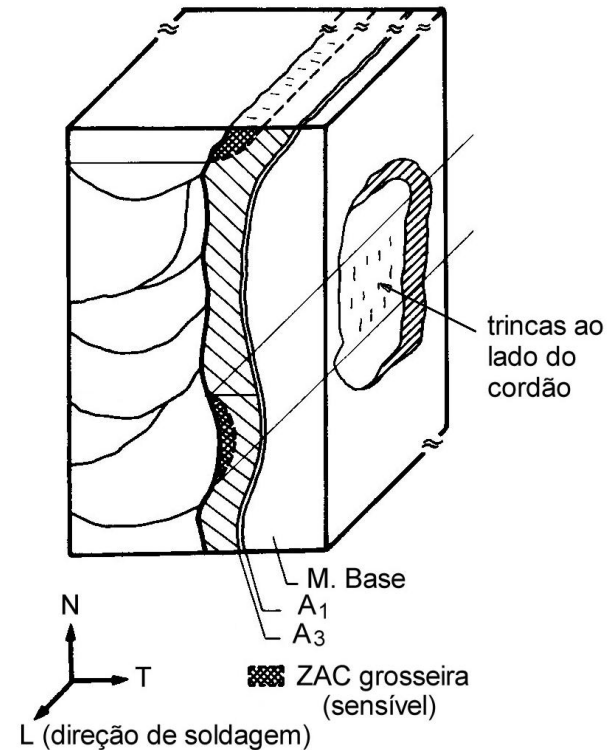
LOCALIZAÇÃO TÍPICA

SOB REVESTIMENTO INOX

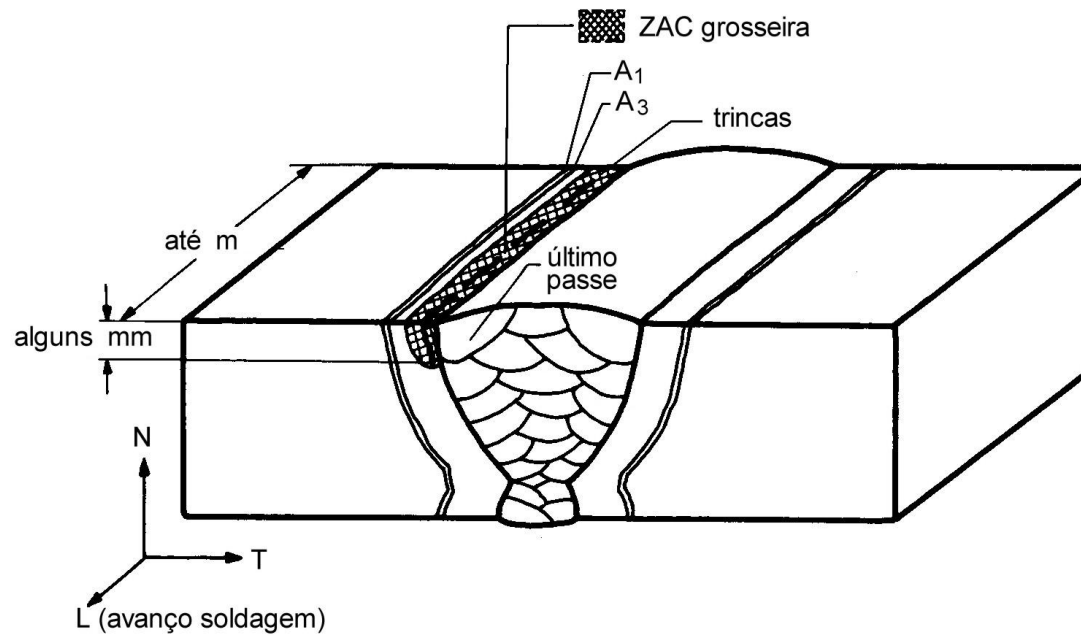


Aço vaso de pressão reatores PWR
22NiCrV37 → 20MnMoNi55

TRINCAS LATERAIS AO CORDÃO

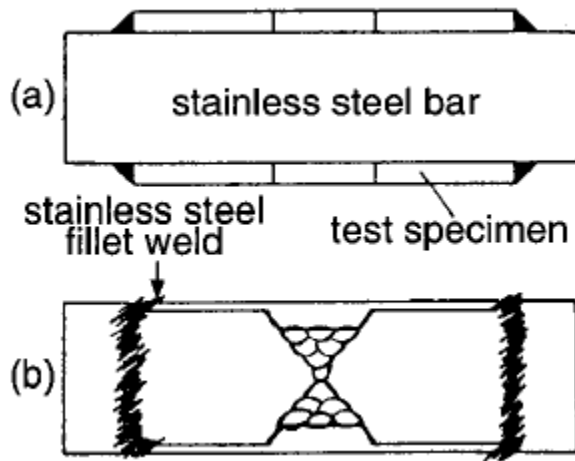


TRINCAS
LONGITUDINAIS AO CORDÃO



Ensaio de Susceptibilidade

Tração a quente, Fluência,
Relaxação, Emissão Acústica etc



Teste de Vinckier

The test specimens are made by welding two pieces of 50-mm-thick plates together. The ends of the test specimens are welded to a stainless steel block. Upon reheating, the test specimens are subjected to tensile loading caused by the higher thermal expansion coefficient of the stainless steel block.

$$\varepsilon_1 = \frac{\sigma_1}{E_1} = \frac{(\alpha_2 - \alpha_1)T}{(E_1 A_1 / E_2 A_2) + 1}$$

where ε is the overall strain in the test specimen, α the thermal expansion coefficient, E Young's modulus, T the reheat temperature, and A the crosssectional area. Subscripts 1 and 2 refer to the test specimen and the stainless steel block, respectively.

MEDIDAS PREVENTIVAS:

a. Quanto ao material

Fórmulas empíricas:

$$CS = \%Cr + 3.3 \times (\%Mo) + 8.1 \times (\%V) - 2$$

Nakamura et al. → aço livre de trincas para $CS < \text{zero}$.

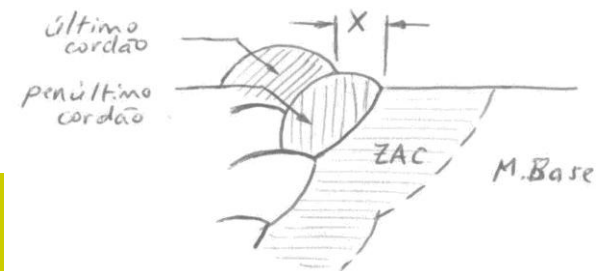
$$Psr = Cr + Cu + 2 Mo + 10 V + 7 Nb + 5 Ti$$

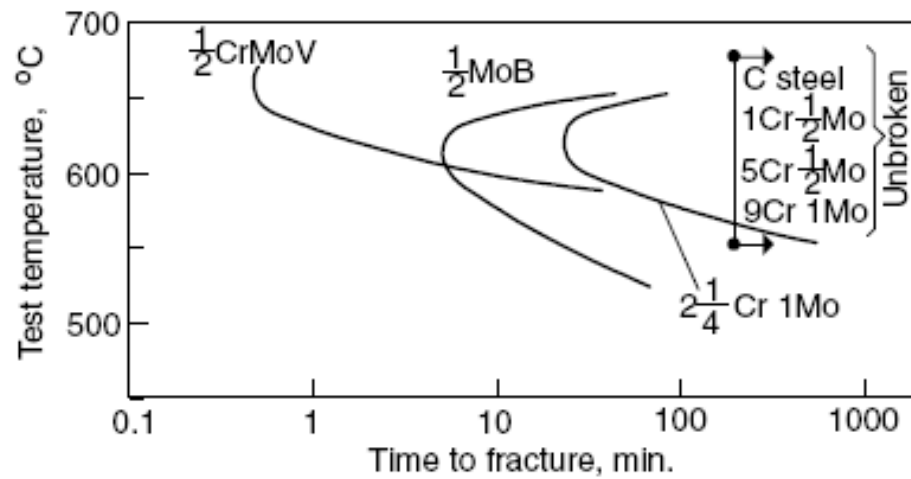
Ito et al. → $Psr < 0$ aço livre de trincas.

22NiMoCr37	→	20 MnMoNi 55
A508	→	A 533
WStE 51	→	15 MnNi 63

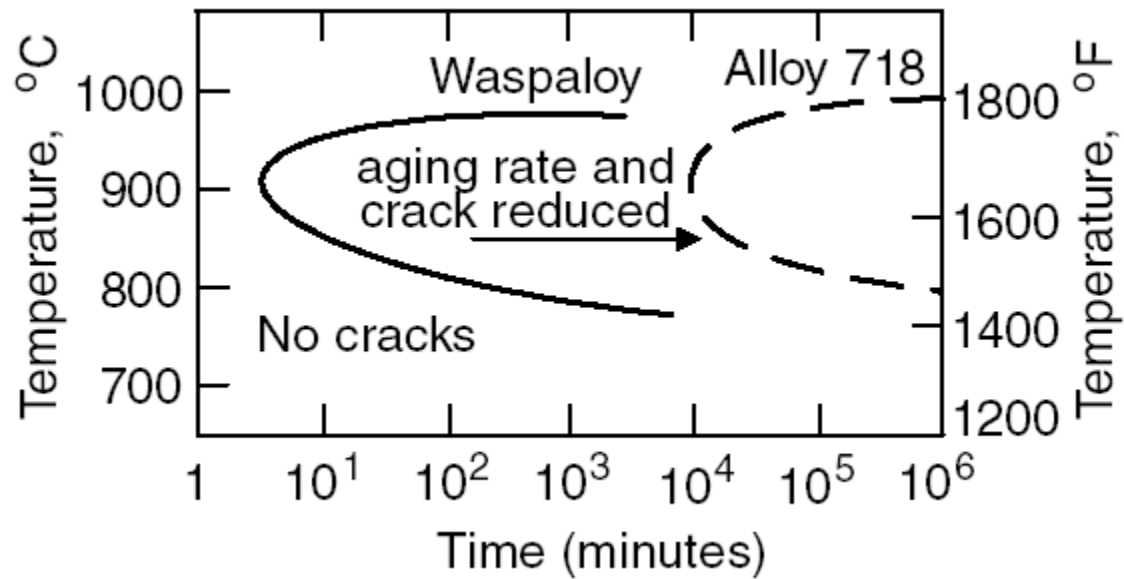
b. Quanto a variáveis de soldagem e ciclo térmico

1. Passe de “revenido” (solução clássica de Vinckier)
2. Soldagem multipasses (visando refino da ZAC-GG)
3. Alívio de tensão (TTPS) por patamares

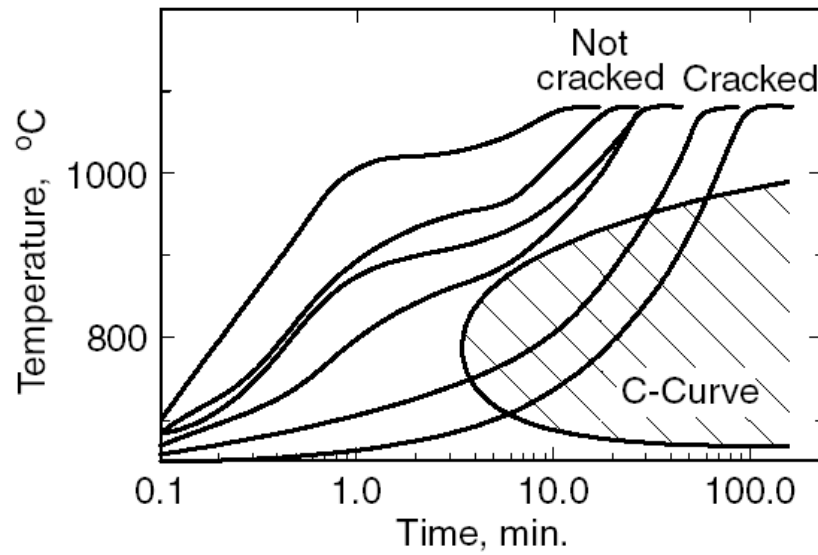




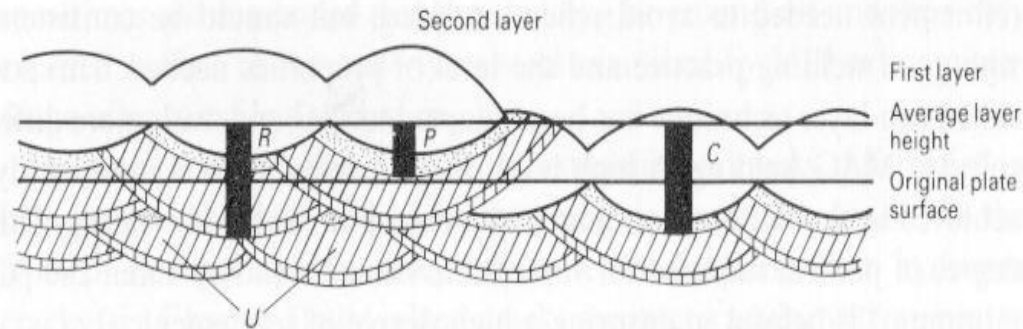
Temperature vs. time to fracture in ferritic steels



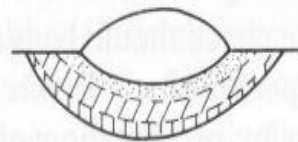
Crack susceptibility C curves for Waspaloy and Inconel 718 welds.



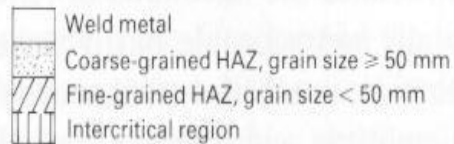
Effect of heating rate on postweld heat treatment cracking of a Rene 41 solution annealed before welding.



Two-layer refinement criterion: $\text{Max. fusion boundary depth, second layer (P)} < \text{max. fusion boundary depth, first layer} + \text{average layer height}, < \text{max. depth of refining zone, second layer (R)}$

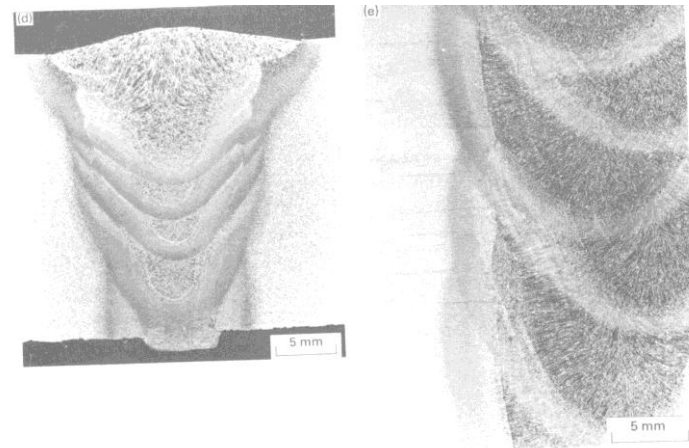
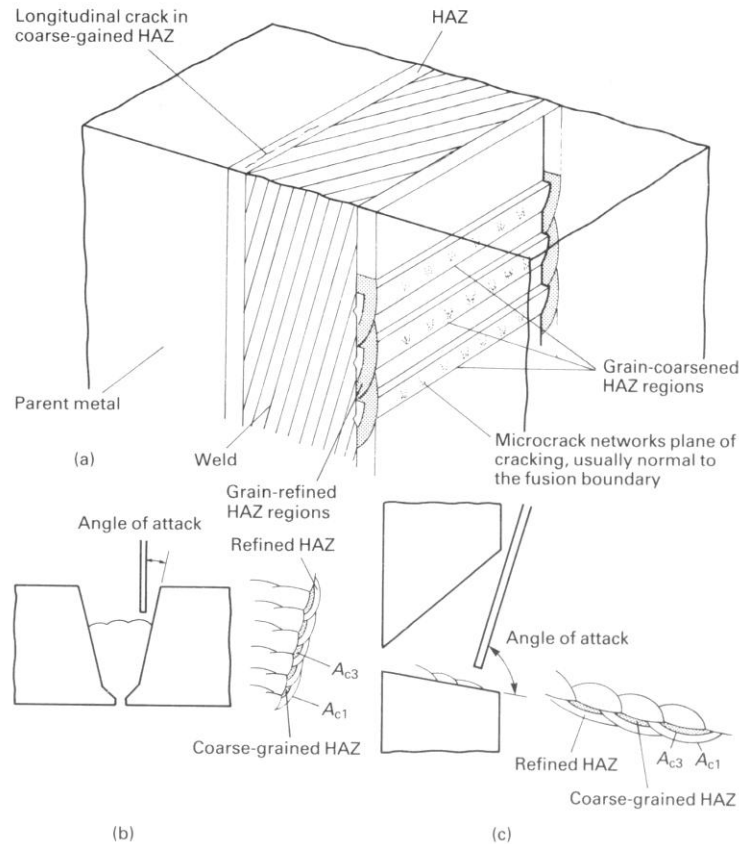


Single weld bead zones

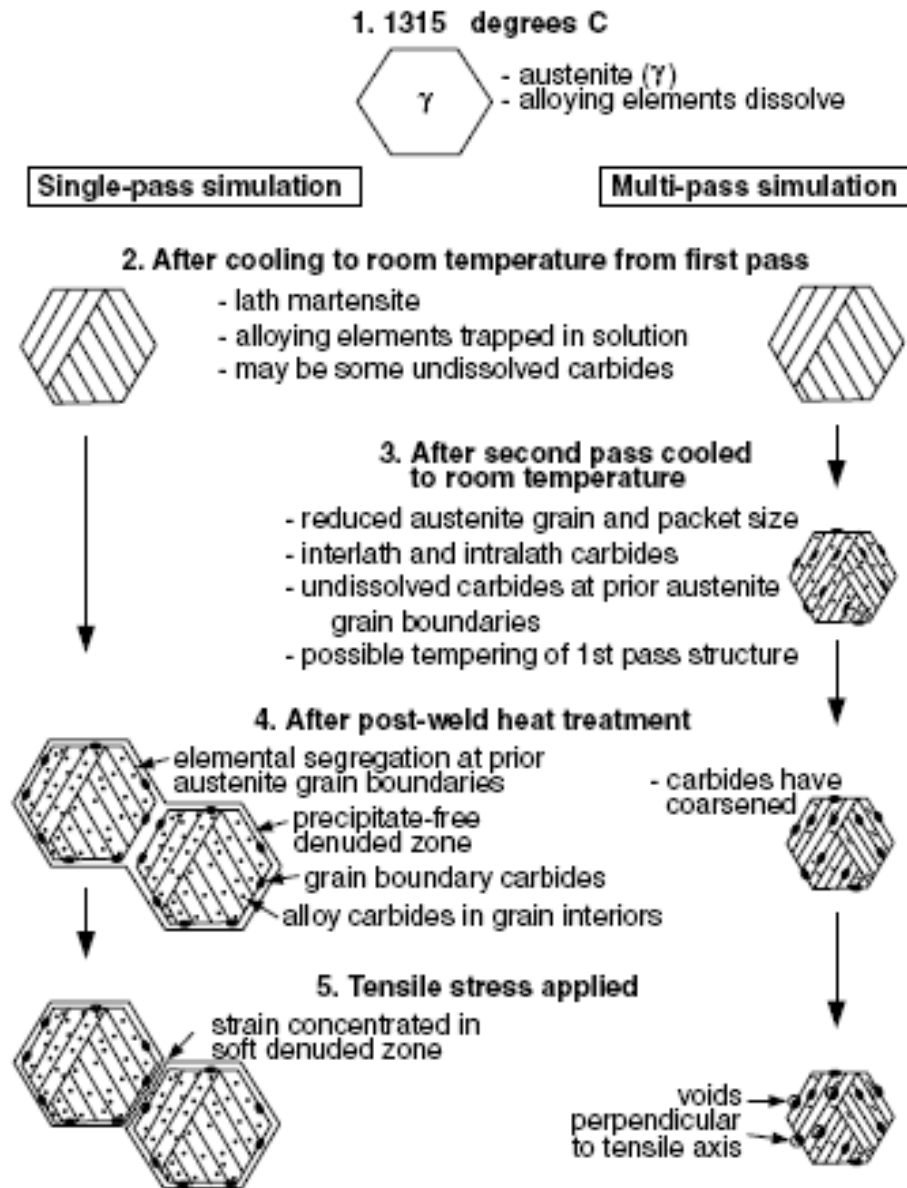


Note:
All measurements with respect to the original surface
The fusion boundary is used as a first order approximation for the depth of the coarse-grained HAZ

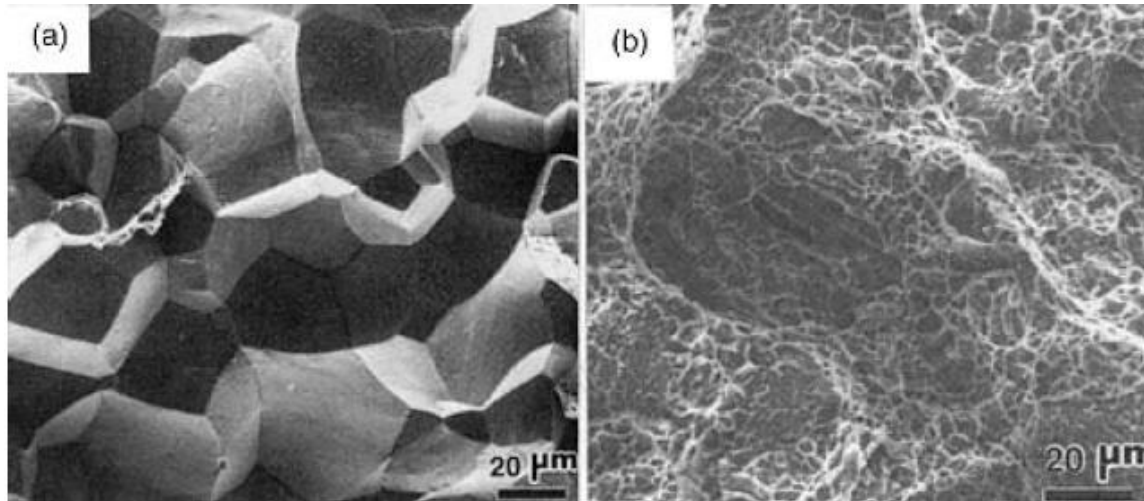
6.3 Sketch showing two-layer technique. Note that the degree of overlap is less than ideal (i.e. $\sim 50\%$ overlap obtained by aiming arc at toe of previous weld bead). 'U' indicates unrefined HAZ regions.



Ângulo de ataque e refino da ZAC: a) local de trincas na ZAC-GG; b) baixo ângulo de ataque e elevada superposição aumentam o refino da ZAC; c) alto ângulo de ataque e baixa superposição diminuem o refino da ZAC; (d) e (e) macroestruturas de soldas com alto e baixo grau de refino da ZAC e superposição, respectivamente.



Microstructural changes and failure mode of single- and multiple-pass samples of a 2.4Cr–1.5W–0.2V ferritic steel.



SEM micrographs of fracture surfaces of a 2.4Cr–1.5W–0.2V ferritic steel: (a) *single-pass sample*; (b) *multiple-pass sample*